A Novel Temperature Stable Microwave Dielectric Ceramic with Garnet Structure: Sr₂NaMg₂V₃O₁₂

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A garnet vanadate ceramic $Sr_2NaMg_2V_3O_{12}$ was prepared by the conventional solid-state route, and the sinterability, microwave dielectric properties, and its chemical compatibility with Ag electrodes were investigated. $Sr_2NaMg_2V_3O_{12}$ sample could be well sintered at $900^{\circ}C$ for 4 h with a relative density of 96.1%. X-ray diffraction data showed that $Sr_2NaMg_2V_3O_{12}$ ceramics crystallized into a cubic garnet structure with a space group Ia-3d over the sintering temperature range $(830^{\circ}C\text{-}930^{\circ}C)$. The $Sr_2NaMg_2V_3O_{12}$ ceramic sintered at $900^{\circ}C$ obtained the optimum microwave dielectric properties with a relative permittivity of 11.7, a $Q\times f$ of 37950 GHz (at 11.0 GHz), and a almost zero τ_f value of -2.9 ppm/ $^{\circ}C$. Chemical compatibility experiments showed no reaction between $Sr_2NaMg_2V_3O_{12}$ ceramics and Ag electrodes.

I. Introduction

OMPOUNDS with the nominal composition $A_3B_2C_3O_{12}$ are known as garnets. In the garnet structure, three different sites (A, B, and C) are available for a wide variety of cations. As illustrated in Fig. 1, C ions are surrounded by four oxygens to form CO_4 tetrahedron; B ions are located in octahedral. The tetrahedra and octahedral are corner shared, forming dodecahedra where A ions are located. Garnet compounds usually crystallize in the cubic system with a space group of Ia-3d. Despite $A_3B_2C_3O_{12}$, garnet-type compounds have been extensively investigated as promising materials in the field of phosphors, lasers, and ferrite materials, $^{6-8}$ there are few reports on the microwave dielectric properties of these compounds.

Kim et al.⁹ firstly reported the microwave dielectric properties of some Re₃Ga₅O₁₂ [(Re=Nd, Sm, Eu, Dy, Yb, and Y) garnets with high quality factors (40 000–192 173 GHz), low relative permittivities (11.5–12.5), and relatively stable temperature coefficients of resonant frequency (–33.7 to –12.4 ppm/°C). However, their sintering temperatures are too high (1350°C–1500°C). In our previous work, some garnet vanadates were reported to have combined low sintering temperatures and good microwave dielectric properties, which make them potential candidates in low-temperature co-fired ceramics (LTCC) technology. ^{10–12} For example, LiCa₃MgV₃O₁₂ ceramic sintered at 900°C has a relative

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permittivity (ϵ_r) ~ 10.5, a quality factor ($Q \times f$) ~ 74 700 GHz, and a temperature coefficient of resonant frequency (τ_f) ~ -61 ppm/°CJ¹⁰ and NaCa₂Mg₂V₃O₁₂ ceramic has a ϵ_r = 10, $Q \times f$ = 50 600 GHz, and τ_f = -47 ppm/°C when sintered at 915°C. ¹¹ Meanwhile, the microwave dielectric properties of some cation-deficient garnets in the Ca₅A₄(VO₄)₆ (A=Mg, Zn, and Co)^{13,14} system were characterized. Researches have shown that these garnet vanadates possessed a combination of low sintering temperatures, relatively high $Q \times f$ values, and low bulk densities, but their large negative τ_f precluded their possible use in practical applications, the search for temperature stable garnets is still going on.

Sr₂NaMg₂V₃O₁₂ belongs to the garnet family. Hu et al.¹⁵ firstly reported the synthesis of Sr₂NaMg₂V₃O₁₂ through solid-state reaction method in detail. Recently, Sr₂NaMg₂V₃O₁₂ has been widely investigated for applications in high-pressure mercury lamp field-emission display and color TV industry^{15–18} due to its luminescent properties. To the best knowledge of us, however, the microwave dielectric properties of Sr₂NaMg₂V₃O₁₂ ceramics have not been investigated. In the present work, the sintering behavior and microwave dielectric properties of Sr₂NaMg₂V₃O₁₂ ceramic were studied. Moreover, the chemical compatibility of Sr₂NaMg₂V₃O₁₂ ceramics with Ag electrode was also investigated.

II. Experimental Procedure

Sr₂NaMg₂V₃O₁₂ ceramics were synthesized by a conventional solid-state method using high-purity oxide powders of Sr₂CO₃ (99%, Guo-Yao Co. Ltd., Shanghai, China), Na₂CO₃ (99%, Guo-Yao Co. Ltd., Shanghai, China), MgO (99%, Guo-Yao Co. Ltd., Shanghai, China), and NH₄VO₃ (>99%, West Long Chemical Co., Ltd., Guangdong, China). MgO was heated at 900°C for 2 h to remove moisture retains. The stoichiometric mixtures of powders were weighed and ball-milled in alcohol medium for 6 h using zirconia balls, followed by the calcination at 800°C for 4 h. The calcined powders were re-milled for 4 h and after drying, the polyvinyl alcohol (PVA, 10 vol%) was added as binders. Then, the powders were pressed into cylinders with 12 mm in diameter and 7 mm in height under a pressure of 200 MPa. The samples were heated to 550°C for 2 h to remove the organic binder and sintered from 830°C to 930°C for 4 h with a heating rate of 5°C/min.

Phase purity of the specimens were studied using an X-ray diffraction measurement ($CuK\alpha I$, 1.54059 Å, Model X'Pert PRO, PANalytical, Almelo, Holland) with $CuK\alpha$ radiation and a monochromator in the 2 θ range of 10°–80°. The apparent densities were measured using the Archimedes method. Microstructures of the samples were observed by scanning electron microscope (FE-SEM, Model S4800; Hitachi, Japan).

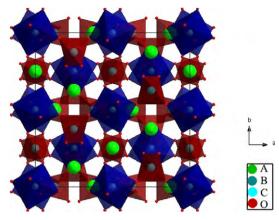


Fig. 1. Crystal structure of A₃B₂C₃O₁₂ garnet compound.

The microwave dielectric properties were analyzed using a network analyzer (Model N5230A, Agilent Co., Palo Alto, California) and a temperature chamber (Delta 9039, Delta Design, San Diego, California). The temperature coefficient of resonant frequency (τ_f) was obtained by noting the temperature variations of the TE₀₁₁ resonance scope from 25°C to 85°C. The τ_f value was calculated as the following relationship:

$$\tau_{\rm f} = \frac{f_{85} - f_{25}}{(85 - 25) \times f_{25}} \tag{1}$$

where, f_{85} and f_{25} were the resonant frequencies of the dielectric resonator at temperature 85°C and 25°C, respectively.

III. Results and Discussions

Figure 2 shows the X-ray diffraction patterns of Sr₂NaMg₂V₃O₁₂ ceramics sintered from 830°C to 930°C. All the sintered Sr₂NaMg₂V₃O₁₂ ceramics showed cubic garnet structure with space group of *Ia-3d* (230) according to the standard JCPDS card No. 024-1131.

The bulk densities of $Sr_2NaMg_2V_3O_{12}$ ceramics as a function of the sintering temperature are presented in Fig. 3. With increasing sintering temperature from $830^{\circ}C$ to $930^{\circ}C$, the bulk density gradually increased, reached a maximum value of $3.73~g/cm^3$ (96.1% of the theoretical density $\sim 3.88~g/cm^3$) at $900^{\circ}C$, and then decreased slightly. SEM image of the $900^{\circ}C$ -sintered sample is shown in the insert of Fig. 3. A uniform and dense microstructure with closely packed grains was observed with average grain size about $3–8~\mu m$.

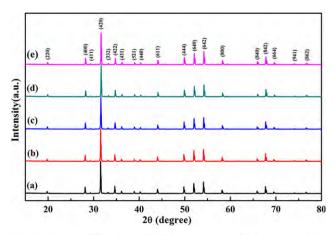


Fig. 2. X-ray diffraction (XRD) patterns of $Sr_2NaMg_2V_3O_{12}$ ceramics sintered at different temperatures: (a) $830^{\circ}C$, (b) $850^{\circ}C$, (c) $880^{\circ}C$, (d) $900^{\circ}C$, and (e) $930^{\circ}C$.

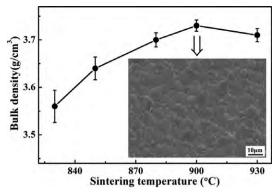


Fig. 3. The bulk densities of the $Sr_2NaMg_2V_3O_{12}$ ceramics as a function of the sintering temperature and the scanning electron microscopy (SEM) image of the fractured surface of ceramic sintered at $900^{\circ}C$.

Figure 4 shows the variation in the relative permittivity (ε_r) , quality factor $(Q \times f)$, and the temperature coefficient of resonant frequency (τ_f) of $Sr_2NaMg_2V_3O_{12}$ ceramics with sintering temperature. ε_r increased from 11.03 to 11.74 as the sintering temperature increased from 830°C to 900°C, and then slightly decreased thereafter. By comparison, it is observed that the variation in the relative permittivity with sintering temperature is similar to that of the density. The largest ε_r value was obtained at which temperature the highest density was achieved. The influence of the porosity on the microwave permittivity could be eliminated by applying Bosman and Havinga's correction 19 as shown in Eq. (2)

$$\varepsilon_{\text{corrected}} = \varepsilon_{\text{m}} (1 + 1.5p)$$
 (2)

where, $\varepsilon_{\rm corrected}$ and $\varepsilon_{\rm m}$ are the corrected and measured values of permittivity, respectively. p is the fractional porosity. The $\varepsilon_{\rm corrected}$ value is about 12.42 for ${\rm Sr_2NaMg_2V_3O_{12}}$ ceramic. Furthermore, ε_r can be interpreted by the sum of ionic polarizability of individual ions (α_D^T) and molar volume $(V_{\rm m})$ according to Clausius–Mossotti equation^{20,21}:

$$\varepsilon_{\rm r} = \frac{\frac{1+2b\alpha_{\rm D}^{\rm T}}{V_{\rm m}}}{\frac{1-b\alpha_{\rm D}^{\rm T}}{V_{\rm m}}} \tag{3}$$

where, $b = 4\pi/3$. The calculated theoretical permittivity of $Sr_2NaMg_2V_3O_{12}$ is 10.5. The relative error of

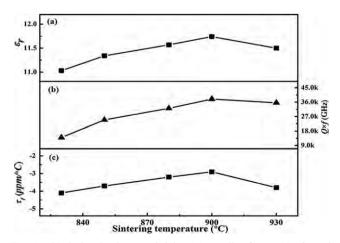


Fig. 4. Variation in the permittivity (ϵ_r) , quality factor $(Q \times f)$, and the temperature coefficient of resonant frequency (τ_f) of $Sr_2NaMg_2V_3O_{12}$ ceramics sintered at different temperatures.

Table I. Comparison of Microwave Dielectric Properties of Some Garnet Vanadates Ceramics

Composition	S.T. (°C)	ϵ_{r}	Q×f (GHz)	$\tau_f \\ (ppm/^{\circ}C)$	Reference
LiCa ₃ MgV ₃ O ₁₂ NaCa ₂ Mg ₂ V ₃ O ₁₂ LiMg ₄ V ₃ O ₁₂ Ca ₅ Mg ₄ (VO ₄) ₆ Ca ₅ Co ₄ (VO ₄) ₆ Sr ₂ NaMg ₂ V ₃ O ₁₂	900 915 740 800 725 875 900	10.5 10 10.7 9.2 11.7 10.6	74 700 50 600 24 000 53 300 49 400 95 200 37 950	-61 -47 -11.7 -50 -83 -63 -2.9	[10] [11] [12] [13] [13] [14] This
5121 varvig2 v 3O12	300	11./4	31 930	-2.9	work

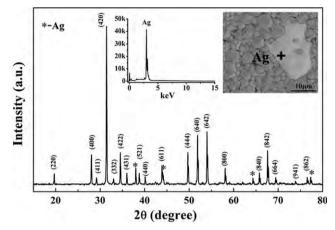


Fig. 5. X-ray diffraction patterns, Backscattered electron image micrograph, and EDS analysis of the Sr₂NaMg₂V₃O₁₂ ceramic with 20 wt% silver powder.

 $Sr_2NaMg_2V_3O_{12}$ is about 11.8% for the measured value and 18.2% for the porosity corrected value, which implies that there is another polarization mechanism in Sr₂NaMg₂V₃O₁₂ ceramic at microwave region beside ionic and electronic displacement polarization.²² Similarly, $Q \times f$ value increased firstly with increasing sintering temperature [as shown in Fig. 4(b)]. At 830°C, a relatively low quality factor ~ 14 000 GHz was obtained. After reaching its maximum (~37 950 GHz) at 900°C, the $Q \times f$ slightly declined to 35 640 GHz at 930°C. Figure 4(c) shows the change in τ_f with increasing sintering temperature. As seen, the τ_f values varied in the range from -4.1 to -2.9 ppm/°C over the sintering region from 830°C to 900°C. A near-zero τ_f value of −2.9 ppm/°C was obtained for sample sintered at 900°C.

Compared with the microwave dielectric properties of some garnet vanadates ceramics (as shown in Table I), it is seen that the sintering temperature and the relative permittivity of Sr₂NaMg₂V₃O₁₂ ceramic are comparable with other garnet vanadates. The most advantage is the thermal stability of $Sr_2NaMg_2V_3O_{12}$ ceramic with a near-zero τ_f value of −2.9 ppm/°C. The mentioned merits make Sr₂NaMg₂V₃O₁₂ a possible candidate in LTCC applications.

To investigate the chemical compatibility of the Sr₂NaMg₂V₃O₁₂ ceramic with silver electrodes, 20 wt% Ag was mixed with Sr₂NaMg₂V₃O₁₂ ceramic and sintered at 900°C for 4 h. XRD pattern, backscattered electron image, and the EDS analysis of the co-fired Sr₂NaMg₂V₃O₁₂ ceramic added with 20 wt% Ag are shown in Fig. 5. From XRD patterns, only the peaks belonging to Sr₂NaMg₂V₃O₁₂ and Ag could be observed without secondary phase detected, indicating no reaction between Sr₂NaMg₂V₃O₁₂ and Ag. This was further confirmed from the EDS analysis, as shown in the inset of Fig. 5. Two distinct grains could be seen. The larger grains marked as "+" were detected to be Ag.

IV. Conclusions

A novel low-temperature stable microwave dielectric ceramic Sr₂NaMg₂V₃O₁₂ with garnet structure was prepared by the conventional solid-state reaction method. The phase purity, packing fraction, and microwave dielectric properties were investigated. Excellent microwave dielectric properties were obtained in Sr₂NaMg₂V₃O₁₂ ceramic sintered at 900°C for 4 h, with a permittivity of 11.74, $Q \times f$ value of 37 950 GHz (at 11.0 GHz), and an almost zero τ_f value of -2.9 ppm/°C. Furthermore, the Sr₂NaMg₂V₃O₁₂ ceramic showed good chemical compatibility with Ag electrode, which makes it a promising candidate for LTCC technology.

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